

Interface study of short mullite fiber reinforced Al-4.5Cu alloy composites

W. LI, H. Y. FAN, X. J. ZHANG

School of Manufacturing Science and Engineering, Sichuan University, Chengdu 610065, People's Republic of China
E-mail: wli@sc.homeway.com.cn

B. L. SHEN, S. J. GAO, M. J. TU

School of Materials Science and Engineering, Sichuan University, Chengdu 610065, People's Republic of China

C. LI, S. Y. QIU

National Key Laboratory of Nuclear Fuel and Materials, Nuclear Power Institute of China, Chengdu 610041, People's Republic of China

The interest in discontinuously reinforced metal matrix composites has increased extensively in recent years. It is possible nowadays to produce many new cheaper aluminum-based composites such as aluminum borate ($\text{Al}_{18}\text{B}_4\text{O}_{44}$) whisker reinforced ones and aluminosilicate ($\text{Al}_2\text{O}_3 \cdot \text{SiO}_2$) short fiber reinforced ones at even lower cost [1–8]. Among various aluminosilicate fibers, short mullite fibers ($3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$) show outstanding properties, which may make this composite a potential candidate for commercial applications in the near future [6–8, 14]. However, many investigators have indicated that there are severe interfacial reactions between the fibers and the matrix in aluminosilicate and/or crystallized aluminosilicate short fiber reinforced aluminum-based composites [9–11], resulting in the formation of MgAl_2O_4 spinel which tends to be detrimental with the attendant loss of fiber strength and leads to decreases in the properties of the composites [12, 13]. There are also researchers who indicate that MgAl_2O_4 spinel was not formed at the interface of mullite short fiber reinforced Al-Si alloy (containing Mg) composite [14]. The question is are there any interfacial reactions in short mullite fiber reinforced aluminum-based composites? The answer is YES based upon our recent research work. The results of the present paper are quite different from those of Cao *et al.* [14] and worth further investigation.

The aluminum alloy matrix used in the present work was an Al-4.5Cu binary alloy, having a chemical composition (in wt%): Cu: 4.45, Fe: 0.23, Mn: 0.04, Zn: 0.01 and the balance aluminum.

Mullite fibers, with a chemical composition (in wt%) $72\text{Al}_2\text{O}_3$ and 28SiO_2 , were selected as the reinforcement. Fig. 1 is the X-ray diffraction ($\text{Cu K}\alpha$, $\lambda = 0.1542$ nm) pattern of the mullite fibers, which is the same as that of [14]. The mullite fibers were chopped and then made into fiber preform as in [15]. The composites were fabricated by squeeze casting with a melt temperature of 1073 K, preform temperature of 723 K, die temperature of 573 K and infiltration pressure of 60 MPa with 2 min of holding during infiltration. The volume fraction of the reinforcing short fibers was about 18%.

The solution treatment involved heating the composites at 788 K for 10 h and then quenching them into ice water. Artificial aging was carried out at 423 K for 54 h.

Specimens for TEM observation were prepared by standard methods involving mechanical grinding, polishing and dimpling followed by ion milling of foils to perforation on a liquid nitrogen-cooled specimen stage to eliminate further aging during the thinning period. Microstructural studies were performed either in a Philips CM12 TEM operating at an accelerating voltage of 100 kV, or in a JEM-200CX TEM at 160 kV, or in a Philips TECNAI 20 at 200 kV.

Fig. 2a is a TEM micrograph of polycrystalline mullite crystals and Fig. 2b shows the $[00\bar{6}]$ electron diffraction pattern from one of the mullite polycrystalline particles. It is clearly shown that the mullite fibers in the composite are polycrystalline made up of $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2$ crystals, which agrees well with the XRD results. Fig. 3 is the HRTEM image of one mullite crystal particle along $[100]$ and, from the micrograph, it was found that the facial distance between (100) planes is about 0.748 nm, which is almost the same as $a = 0.749$ nm [16]. Brandes [16] tells us that

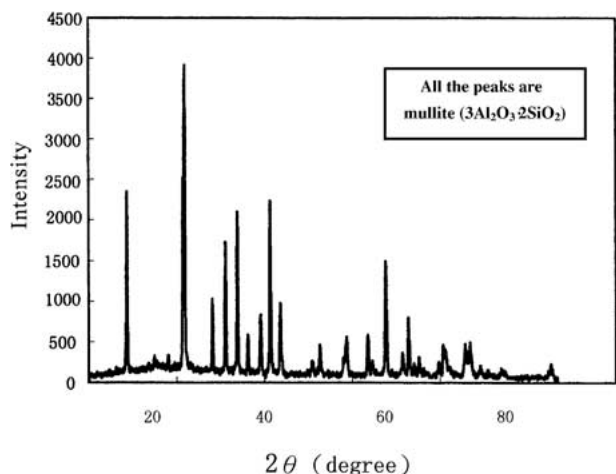


Figure 1 X-ray diffraction (XRD) pattern of the mullite fibers.

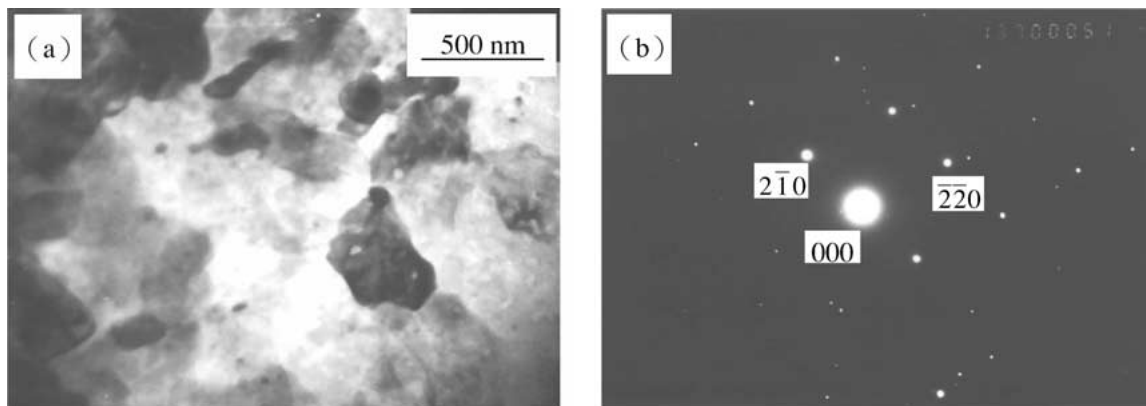


Figure 2 TEM images of polycrystalline mullite (a) and electron diffraction pattern of one crystal mullite particle (b).

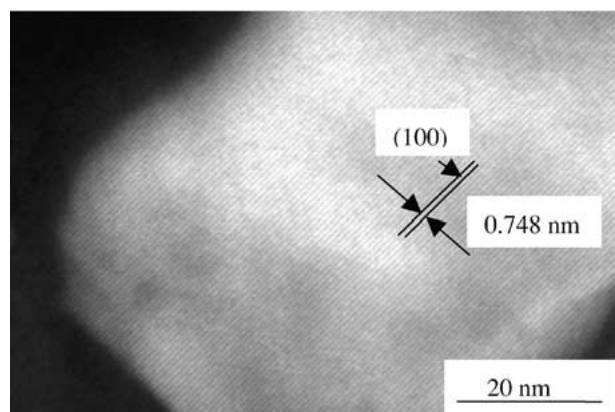


Figure 3 HRTEM image (DF) of one crystal mullite particle along [100].

mullite is orthorhombic in structure with lattice constants: $a = 0.749$ nm, $b = 0.927$ nm and $c = 0.581$ nm. The result of the present study is different from that of [14] in which mullite is taken as tetragonal in structure with lattice constants: $a = 0.755$ nm, $b = 0.769$ nm and $c = 0.288$ nm. Fig. 4a is a TEM micrograph of a typical interface in $(3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2)/\text{Al}-4.5\text{Cu}$ composites. It can be seen from Fig. 4a that there exist many aged θ' (Al_2Cu) precipitates in the Al-4.5Cu matrix and also a few interfacial reaction products at the mullite fiber/matrix interface. Fig. 4b is the electron diffraction pattern of one of the reaction products, which has a cubic type structure with a lattice constant $a = 0.808$ nm. This product can be identified as CuAl_2O_4 [16].

In [14], based on the phase diagram of the Al-Si binary system and the TEM micrograph of the mullite fiber/Al-Si matrix interface, the authors deduced that the Mg_2Si particle at the interface is an eutectic phase, not a reaction product between the mullite fiber and the matrix alloy. They explained further that the main difference between the aluminosilicate fibers and the mullite fibers is the lower silica (SiO_2) content (28 wt%) in mullite fibers and that the interaction time (20 s for solidification) between the silica in mullite fibers and magnesium in matrix alloy is limited. So, there could not be sufficient silica and time for spinel (MgAl_2O_4) formation.

Unlike the case of [14], there is a reaction product CuAl_2O_4 formed at the interface of mullite fiber reinforced Al-4.5Cu alloy composites. The phase diagram of the Al-Cu binary alloy system shows that Al_2Cu is a eutectic phase. In as-cast conditions, it is easy to observe the non-equilibrium eutectic Al_2Cu phase. After solution treatment of heating at 788 K for 10 h, followed by ice water quenching, the Al_2Cu phase might re-melt into the matrix, resulting in much fewer precipitates of Al_2Cu at the fiber/matrix interface. It is possible that some eutectic Al_2Cu phase may still exist at the fiber/matrix interface. Even in this situation, it is not difficult to identify the Al_2Cu phase from the CuAl_2O_4 spinel.

From the above observations and discussion, we can conclude that chopped mullite fibers are polycrystalline in structure composed of many fine mullite

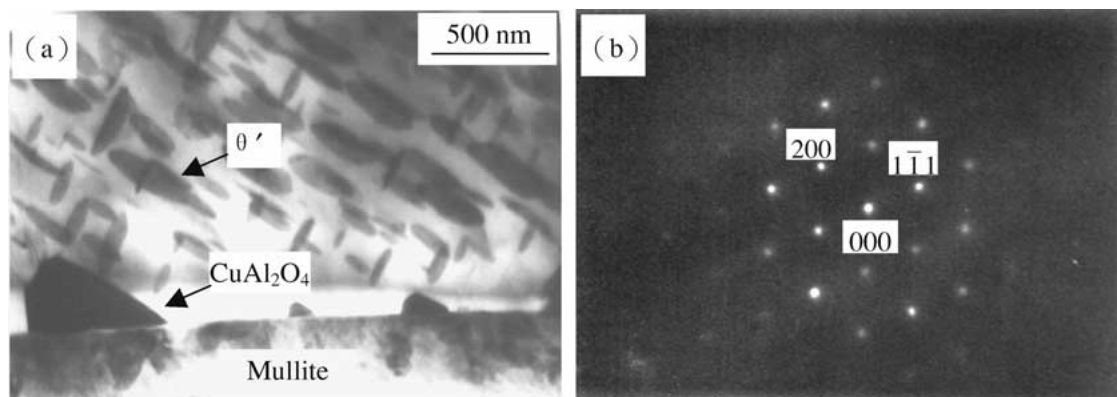


Figure 4 TEM image of a typical $3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2/\text{Al}-4.5\text{Cu}$ interface (a) and electron diffraction pattern of one of the interface reaction products (b).

crystal particles. Mullite is orthorhombic with lattice constants of $a = 0.749$ nm, $b = 0.927$ nm and $c = 0.581$ nm based on electron diffraction analysis and HRTEM observation of crystal mullite. After solution and aging treatment, there appeared a reaction product CuAl_2O_4 formed at the fiber/matrix interface in $(3\text{Al}_2\text{O}_3 \cdot 2\text{SiO}_2)/\text{Al}-4.5\text{Cu}$ composites.

References

1. N. NISHINO and S. TOWATA, *J. Japanese Ins. Light Met.* **47** (1997) 208.
2. L. J. YAO and H. FUKUNAGA, *Scripta Mater.* **36** (1997) 1267.
3. L. J. YAO, G. SASAKI, J. PAN, M. YOSHIDA and H. FUKUNAGA, *Metall. Mater. Trans. A* **29** (1998) 1517.
4. S. KAVECKY and P. SEBO, *J. Mater. Sci.* **31** (1996) 757.
5. D. NATH and V. SINGH, *Scripta Mater.* **40** (1999) 791.
6. S. CANUMALLA, S. A. DYNAN, D. J. GREEN, R. B. BKAGAT and R. N. PANGBORN, *J. Comp. Mater.* **29** (1995) 653.
7. W. LI, B. L. SHEN, S. JING, S. J. GAO, M. J. TU, H. H. WANG, Z. G. WU, D. ZHANG and R. J. WU, *The Chinese J. Nonferr. Met.* **11** (2001) 253.
8. S. J. CHU and R. J. WU, *Acta Mater. Comp. Sinica* **15** (1998) 33.
9. S. Q. WU, H. Z. WANG and S. C. TJONG, *Comp. Sci. Tech.* **56** (1996) 1261.
10. G. H. CAO, S. Q. WU, G. J. SHEN and G. J. SHU, *J. Mater. Sci. Lett.* **18** (1999) 393.
11. G. H. CAO, G. J. SHEN, J. M. LIU and Z. G. LIU, *ibid.* **19** (2000) 1311.
12. M. WANG, Z. WANG and G. C. WEATHERLY, *Mater. Trans.* **23A** (1992) 1423.
13. F. U. REHMAN, S. FOX, H. M. FLOWER and D. R. F. WEST, *J. Mater. Sci.* **29** (1994) 1636.
14. G. H. CAO, G. J. SHEN, Z. G. LIU and S. Q. WU, *J. Mater. Sci. Lett.* **20** (2001) 501.
15. S. C. TJONG, H. Z. WANG and S. Q. WU, *Metall. Mater. Trans.* **27A** (1996) 2385.
16. E. A. BRANDES, "Smithell's Metals Reference Book" 6th edn. (Betterworth, London, 1983).

Received 10 March
and accepted 26 September 2003